

**МАТЕРИАЛЫ И ПРОТИВОКОРРОЗИОННЫЕ РАЗРАБОТКИ В МОРСКОЙ  
И ПОДВОДНОЙ ДОБЫЧЕ НЕФТИ И ГАЗА****М.И. Сизяков**

**Михаил Игоревич Сизяков** (ORCID 0000-0001-6065-6392), ООО «Луховицкая нефтебаза», ул. Советская, 36-А, Луховицы, Московская обл., Российская Федерация, 140501

**Область научных интересов:** проектирование, обслуживание и эксплуатация газонефтепроводов и газонефтехранилищ.

**Mikhail I. Sizyakov** (ORCID 0000-0001-6065-6392), LLC "Lukhovitskaya oil depot", Sovetskaya st., 36-A, Lukhovitsy, Moscow reg., 140501, Russia

**Research interests:** design, maintenance and operation of gas and oil pipelines and gas and oil storages.

E-mail: mixsiz@spbu.su

*Освоение углеводородных ресурсов континентального шельфа Российской Федерации и особенно его арктической части и Дальнего Востока, является крупнейшим инфраструктурным проектом, основанным на уникальных разведанных запасах нефти и газа. Континентальный шельф России является крупным национальным запасом углеводородов, который может обеспечить до 25% российской нефти и до 30% российского газа. Наиболее сложные запасы представляют собой сложные условия разработки – высокое давление и низкие температуры, могут находиться в глубоководных или арктических регионах. Современная добыча нефти осуществляется, зачастую, в экстремальных условиях: при высоких давлениях, в агрессивных морских средах и при низких температурах. Комплексный метод разработки месторождений позволяет длительное время поддерживать высокий уровень добычи углеводородов, снижая затраты за счет оптимизации используемых ресурсов и использования общей инфраструктуры. В перспективе наличие производственных мощностей и сервисных баз вблизи арктического шельфа позволит решать мультидисциплинарные задачи, затрагивающие различные сегменты и направления, такие как инженерно-технологический, природоохранный и другие. В данной статье рассматриваются факторы, влияющие на стойкость нефтегазопроводов к коррозии в морской среде. Приведен обзор методов упрочнения и легирования коррозионностойких сплавов для эксплуатации в морской и подводной среде при расширении границ применения традиционных технических сплавов. Проанализированы направления дальнейшего изучения механизмов коррозии и распространения трещин, которые могут привести к разрушению трубопроводов. Исследование наномасштабной коррозии, проанализированное в этой статье, может оказать глубокое влияние на характеристики деградации материалов, применимых ко всей отрасли.*

**Ключевые слова:** морские трубопроводы, коррозия, защита от коррозии, коррозионностойкие сплавы

**Для цитирования:**

Сизяков М.И. Материалы и противокоррозионные разработки в морской и подводной добыче нефти и газа. *Изв. вузов. Химия и хим. технология.* 2023. Т. 66. Вып. 4. С. 6–16. DOI: 10.6060/ivkkt.20236604.6739.

**For citation:**

Sizyakov M.I. Materials and anticorrosion developments in offshore and subsea oil and gas production. *ChemChemTech* [Izv. Vyssh. Uchebn. Zaved. Khim. Khim. Tekhnol.]. 2023. V. 66. N 4. P. 6–16. DOI: 10.6060/ivkkt.20236604.6739.

## MATERIALS AND ANTICORROSION DEVELOPMENTS IN OFFSHORE AND SUBSEA OIL AND GAS PRODUCTION

M.I. Sizyakov

*The development of hydrocarbon resources of the continental shelf of the Russian Federation and especially its Arctic part and the Far East is the largest infrastructure project based on unique proven oil and gas reserves. The continental shelf of Russia is a large national hydrocarbon reserve, which can provide up to 25% of Russian oil and up to 30% of Russian gas. The most difficult reserves represent difficult development conditions - high pressure and low temperatures, may be in deep water or arctic regions. As oil resources become more and more heterogeneous, an expansion of various methods of extraction and processing is required, including in extreme conditions, at high pressures in aggressive marine environments and low temperatures. An integrated field development method allows maintaining a high level of hydrocarbon production for a long time, reducing costs by optimizing the resources used and using a common infrastructure. In the long term, the presence of production facilities and service bases near the Arctic shelf will allow solving multidisciplinary tasks that affect various segments and areas, such as engineering and technology, environmental protection, and others. The review article discusses the factors affecting the resistance of oil and gas pipelines to corrosion in the marine environment, the issues of hardening and alloying corrosion-resistant alloys for use in the marine and underwater environment while expanding the boundaries of the use of traditional technical alloys, as well as further study of the mechanisms of corrosion and crack propagation, which can lead to equipment failure and destruction. The study of nanoscale corrosion analyzed in this article can have a profound impact on the degradation characteristics of materials applicable to the entire industry.*

**Key words:** offshore pipelines, corrosion, corrosion protection, corrosion-resistant alloys

### INTRODUCTION

Due to the depletion of reserves of traditional oil and gas fields over the past four decades, the industry has moved to the development of more complex fields [1-5] in condition of high pressure and low temperatures in Arctic regions [6, 7].

Thus, the uninterrupted oil production has been carried out on the Russian Arctic shelf for more than 5 years. The continental shelf of Russia is a large national hydrocarbon reserve, which can provide up to 25% of Russian oil and up to 30% of Russian gas [8]. According to the refined results of a quantitative assessment of hydrocarbon (HC) resources, it has been established that reserves of natural gas, condensate, oil and dissolved gas in the amount of more than 122 billion tons of fuel equivalent are concentrated on the shelves of the seas of Russia [9].

Such projects include, for example, Prirazlomnaya and Sakhalin-1. Prirazlomnaya is the first and only Russian project on the Arctic shelf, providing a full production cycle (drilling, production, processing,

storage, offloading to tankers). At the peak, it is planned to produce 4.8 million tons of oil equivalent per year in 2022 [9, 10]. Sakhalin-1 is another large project in Russia for the development of hydrocarbon reserves in subarctic conditions; the balanced revenues to the budget of the Russian Federation amount more than 1.2 trillion rubles [11].

In 2014-2020 various sectoral sanctions were introduced, limiting foreign financing of leading state-owned banks, oil and gas companies and Russian oil and gas limited liability companies, access to advanced production technologies [12].

The government introduced an import substitution policy to localize the production of materials and stimulate the development of innovative technologies for the oil and gas sector in order to reduce dependence on imported technologies, as well as to attract foreign investment for the development of high-strength materials, including carbonaceous, as well as corrosion-resistant alloys associated with the production of hydrocarbons in arctic regions. Therefore, great interest has arisen in the study of corrosion-resistant alloys.

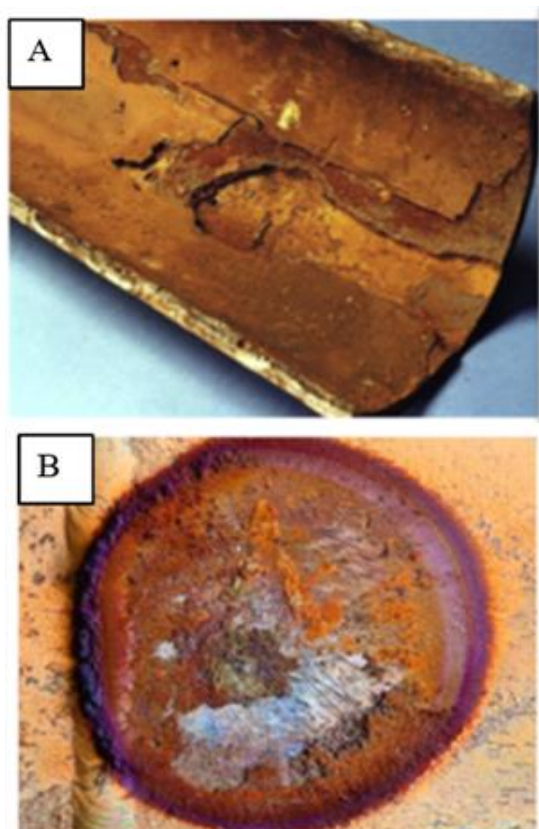


Fig. 1. Common forms of corrosion in transmission oil and gas pipelines: A) internal corrosion, B) pitting [18]

Рис. 1. Распространенные формы коррозии на трансмиссионных нефте- и газопроводах: А) внутренняя коррозия, В) точечная коррозия [18]

There are many review articles on this topic, one of the latest, published in 2020, on the patterns of internal corrosion and anti-corrosion protection of offshore facilities [13]. This review is devoted to environmental cracking of high-strength alloys, methods of hardening and alloying of corrosion-resistant alloys, advanced methods of studying corrosion, such as studying micro- and nanocracks at grain boundaries.

Materials used in oil and gas production in the Arctic regions are exposed to the most aggressive environments. Although the number of serious incidents in the oil and gas industry is not alarming, corrosion of materials can lead to costly catastrophic failures with serious consequences for human life and the environment [14, 15].

This review article discusses the major materials science challenges faced in the oil and gas industry and demonstrates the importance of industry and research synergies.

#### PREVALENT FORMS OF PIPELINE CORROSION

The Russian Federation as an energy power has an important competitive advantage – a developed

and constantly expanding network for the delivery of energy resources [16]. The integrity of pipelines transporting and distributing oil, gas, petroleum products, and other substances is seriously threatened due to electrochemical deterioration (so-called corrosion) [17, 18].

Pipeline corrosion is the deterioration of the material of pipes and associated systems due to their interaction with the service environment. Corrosion of the pipeline and the resulting failures, as well as possible repairs and monitoring costs annually cost the global economy billions of dollars [18]. Corrosion affects all buried or submerged oil and gas pipelines, as they are usually made of metal - mostly steel, with the exception of components and assembly lines.

Corrosion of pipelines occurs due to an electrochemical reaction in the presence of an electrolyte in an aqueous medium; this is usually soil water or fractions of the products that they transport (Fig. 1). Electronic transfer is a very important component of the corrosion process. Monitoring and mitigation systems rely on monitoring the voltages and currents associated with the corrosion process [19].

#### FACTORS AFFECTING THE RESISTANCE OF OIL AND GAS PIPES TO CORROSION IN THE MARINE ENVIRONMENT

##### *Hydrogen sulfide stress corrosion cracking (HSSCC)*

Stress corrosion cracking is the growth of cracking in an aggressive environment. It can lead to unexpected, sudden fracture of normally ductile metal alloys subjected to tensile stress, especially at elevated temperatures [20, 21]. Stress corrosion cracking has a high chemical specificity, as some alloys can only undergo it when exposed to small amounts of chemical environments. The chemical environment that causes cracking for a given alloy is often the environment that causes only minor metal corrosion [22, 23]. Consequently, metal parts with severe stress corrosion cracking can appear bright and shiny while filled with microscopic cracks. Stress corrosion cracking progresses rapidly and is more common among alloys than pure metals [24].

The experience of operating various oil and gas equipment has shown that low- and medium-strength steels with a yield point not exceeding 560 MPa have a sufficiently satisfactory resistance to HSSCC. Earlier, abroad in the oil refining industry, steels with yield strength of 280-390 MPa were successfully used for the manufacture of equipment. However, when using high-strength materials with a yield point of at least 740 MPa, the problem of corrosion destruction of

equipment in the presence of hydrogen sulfide arose, which was associated with an increase in the depth of drilled wells.

With an increase in the strength of steel, its resistance to hydrogen sulfide stress corrosion cracking sharply decreases [25-27]. Fig. 2 shows the dependence of the threshold stress, at which no cracking of structural and pipe steels occurs, on the yield strength. With increasing hardness, the tendency to HSSCC increases (the time to fracture decreases). The resistance of steels with the same hardness decreases with an increase in the content of hydrogen sulfide in the medium [28, 29].

As the strength of the steel increases, the probability of fracture along the grain boundaries increases. With an increase in the yield point from 725 to 1210 MPa, the nature of the destruction of steels in an atmosphere of hydrogen sulfide at a partial pressure from 0.13 to 0.44 MPa changes from transcrystalline to intercrystalline [30].

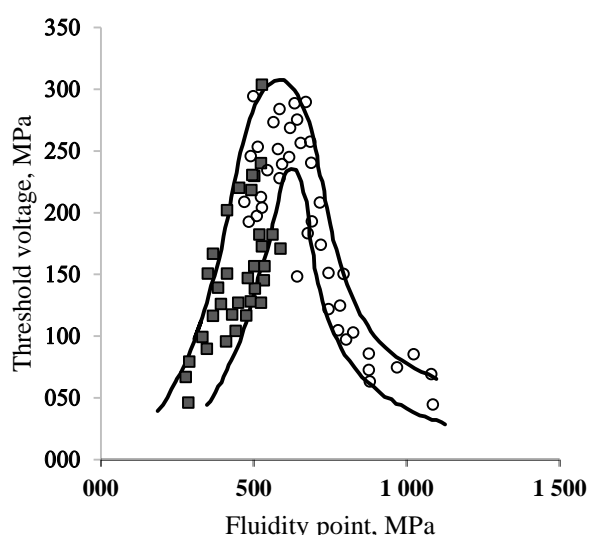


Fig. 2. Resistance to hydrogen sulfide stress corrosion cracking of welded structural manganese (■) and improved pipe (○) steels for oil pipelines, depending on the yield strength [27]. The curves correspond to the boundaries of the scatter of values

Рис. 2. Стойкость к сероводородному коррозионному растрескиванию под напряжением свариваемых конструктивных марганцовистых (■) и улучшенных трубных (○) сталей для нефтепроводов в зависимости от предела текучести [27]. Кривые соответствуют границам разброса значений

Thus, carbon and low-alloy steels with  $\sigma_{0.2}$  not more than 650 MPa, which corresponds to HRC 22 hardness according to Rockwell, can undergo HSSCC under especially severe conditions. In this case, to increase the resistance to cracking, one should not go to steels with a lower  $\sigma_{0.2}$ , but it is necessary to reduce tensile stresses.

### *Influence of the chemical composition of steel.*

#### *Low alloy steels*

Steel is a multi-phase material consisting of iron (ferrite) and iron carbide  $\text{Fe}_3\text{C}$  (cementite). Cementite is more chemically stable than ferrite and does not dissolve during carbon dioxide corrosion.

So, the standard nominal composition of typical carbon and low-alloy steels includes, according to the requirements of ASTM A508, from 1.5-2.0 wt.% Chromium (Cr), from 0.4-0.6 wt.% Molybdenum (Mo), from 2.8-3.9 wt.% nickel (Ni), carbon (C) content – 0.23 wt%, silicon (Si) – 0.4 wt% [15]. The standard minimum value of the yield strength for such steel is 450-690 MPa.

High strength materials with increased fatigue life are required to overcome the design challenges associated with extreme well pressures and low temperatures in arctic regions. Hydrogen cracking resistance decreases with increasing strength [31]. Thus, there is an upper limit for the safe use of technical alloys in oil and gas production environments, which is arguably more conservative than in other industries [32].

There is no universal definition of what constitutes a high strength material. In the context of this paper, high strength refers to materials with specified minimum tensile strength values above the typical maximum currently recommended for forged carbon and low alloy steels exposed to operating fluids, i.e. 550-586 MPa.

For martensitic and sediment-hardened martensitic foreign stainless-steel grades, such as UNS S41000, UNS S17400, Cr content – 11.5-13 wt%, Ni from 3.0-5.0 wt%, these grades may contain niobium (Nb) – from 0.15 to 0.45 wt%, Carbon (C) – 0.15 wt% [15].

Russian scientists also present their experience in the production of high-strength casing and tubing from corrosion-resistant martensitic steels with a Cr content of 13 wt% [33] with a wide range of strength characteristics, including for achieving minimum yield strengths of 552-758 MPa in pipes in conventional and cold-resistant versions.

Bench and field tests of pipes and developed threaded connections have confirmed the compliance of products with the stated requirements. Based on the test results, the products are used at the facilities of the Gazprom Group.

Exploration and production of oil and gas is moving to the Arctic regions [34]. Components operating in arctic conditions can be exposed to extremely low temperatures, which requires materials and welds that maintain high strength and fatigue characteristics down to  $-60\text{ }^\circ\text{C}$  [35, 36].

Low alloy steels (LAS) are among the most advanced engineering materials. In terms of volume, the use of LAS in various areas of the oil and gas industry far exceeds the use of any other family of alloys [37]. Consequently, improving the properties and productivity of the LAS can have a significant impact on the development of oil fields in difficult conditions.

In spite of their advantages, LASs nevertheless undergo environmental corrosion, for example, in environments containing H<sub>2</sub>S, and due to hydrogen generated by cathodic protection systems [38, 39]. The intensity of corrosion of underground metal structures mainly depends on the composition of the soil, its electrical resistivity, the presence of water, oxygen, etc. in the soil [40, 41].

Nowadays, most low carbon steels are accepted for use in H<sub>2</sub>S environments if they contain 1 wt% Ni and the hardness of the surface exposed to the hydraulic fluid is kept below 250HV. For example, quenched and tempered mild steels with strength values below 550 MPa are believed to withstand exposure to H<sub>2</sub>S at stresses up to 100% of their actual yield strength at a total pressure of 1 atm [42].

Data collected by Kappes et al. [43] show that hardened and tempered steels and bainitic steels are the most resistant to sulfide stress cracking. Unhardened steels containing fresh martensite are highly susceptible to hydrogen attack [44, 45].

Researchers have recently developed alloys with yield strengths up to 860 MPa. These materials resist sulfide cracking in mild to moderately acidic operating conditions [46] due to advances in grain boundary development [47-49]. The authors of these works found that the high dissipation energy of special high-angle grain boundaries, more than 30 °, reduces the driving force for crack propagation.

3.3 Hardening and alloying of corrosion resistant alloys for marine and subsea applications

Typically, large diameter subsea field components such as valves, connectors and pipes are made of carbon steel lined with corrosion resistant alloys. In subsea oil and gas production, carbon steel is usually deposited with nickel-containing alloys [50].

Both stainless steel and nickel alloys find numerous applications in the oil and gas industry. In particular, nickel-based alloys are widely used in wellbore components due to their combination of strength and resistance to stress corrosion cracking [51-53]. The most common nickel alloys, for example, UNS N07718 (NA718), contain 17-21 wt% Cr, 2.8-3.3 wt% Mo, 50-55 wt% Ni, Nb, Ta, and Ti [54]. Despite its excellent performance in acidic industrial environ-

ments, NA718 is susceptible to pitting and crevice corrosion in oxidizing halogen environments due to the intermediate Cr and Mo content.

There are also other nickel alloys that can withstand the most corrosive acidic environments and are considered to be resistant to seawater [55]. Currently, no standard defines the maximum allowable temperature for seawater service for such alloys; however, according to ISO 21457 the limitation is 30 °C due to crevice corrosion problems in chlorinated systems.

A priori, nickel alloys were considered immune to hydrogen embrittlement under conditions of increased strength [56, 57]. However, during the installation and operation of the equipment, sudden failures and cracks of subsea structural steel components have been reported under relatively favorable conditions associated with hydrogen embrittlement.

A variety of corrosion resistant alloys are used in the oil field, including martensitic, austenitic, ferritic, duplex and stainless steels, annealed and nickel alloys, and titanium, cobalt and aluminum alloys. This number of alloys is required to handle stress corrosion cracking, sulfide cracking, and galvanic induced hydrogen cracking. Standards (ISO, GOST) limit strength and hardness in some alloy systems. The material boundaries established by the standard are derived from a combination of industry experience and qualification testing.

#### INFLUENCE OF HYDROGEN ON LOCALIZED CORROSION RESISTANCE OF CORROSION-RESISTANT ALLOYS. STUDY OF HYDROGEN-DISLOCATION INTERACTIONS

Hydrogen is the smallest atom in the universe, and the presence of H leads to severe deterioration in strength and toughness. Recent simulations show that the small size of the H atom in the crystal lattice leads to the formation of asymmetric bonds between the H atoms and the host metal [58].

Some authors [59-61] have shown that the hydrogen present in the passive film reduces the resistance to pitting corrosion due to its strong reducing properties.

Thus, a typical, but not trivial, approach is to reduce the sample size and perform micro- and nanoscale mechanical estimates of crack propagation [62, 63].

Undoubtedly, in recent decades, nanoindentation has been the most popular and frequently used small-scale testing method [64].

A typical nanoindentation test consists of several stages. The first one is to obtain an image of the surface relief, then the tip can be positioned with nanometer accuracy. Subsequently, multiple punching of



the material can be performed. In such cases, indentation begins with an elastic load that follows the Hertzian contact model [65].

When the shear stress below the tip in the bulk of the material approaches the theoretical stress required for the nucleation of a homogeneous dislocation, a sudden jump occurs. Then indentation continues in elastoplastic mode up to the maximum indentation load. It can be assumed that the unloading curve is completely elastic and is usually used to determine the hardness and elastic modulus of a material by the Oliver-Pharr method [66].

Nanoindentation provides excellent opportunities for studying the effect of hydrogen on mechanical properties, especially the effect of hydrogen on dislocation nucleation. The required load for the nucleation of a homogeneous dislocation decreases in the presence of H; the dislocation nucleates more easily [67].

#### LOCALIZED CORROSION OF CARBON STEEL AT THE MICRO LEVEL IN REAL TIME

Pitting corrosion in steel is initiated at or near inclusions within the microstructure, such as MnS and iron carbides [68]. However, it is not known why some inclusions, even of the same composition, are more electrochemically active than others [69]. There are three theories explaining this mechanism: (1) orientation of the surface of the iron matrix, (2) galvanic mediation, [69] or (3) disordered and stressed iron matrix [70, 71].

Inclusions of iron carbide, in particular cementite ( $\text{Fe}_3\text{C}$ ), are of particular interest in the study of corrosion of carbon steel. Experimental observations of nanoscale solid-liquid interfacial processes are limited by the complexity of reproducing the flowing corrosive medium inside a device capable of characterizing rare, stochastic, and corrosion-initiating events occurring at the nanoscale [72].

Some authors [73] studied steel corrosion processes using TEM methods to obtain a temporary object-specific nanoscale visualization of localized corrosion processes occurring simultaneously on many different solid-liquid interfaces present in a pipeline sample from the real world.

The fully characterized sample was placed in a liquid chamber for real-time observation under water flow. The corroded sample was then characterized using TEM methods.

In one area near the center of the steel specimen, clear signs of localized accelerated corrosion were found. The first signs of localized corrosion were visible after 40 min of exposure to liquid electrolyte, as indicated by the rapid changes in intensity in the TEM

micrographs. Comparison of these brighter areas with preliminary structural data showed that the initiation of corrosion occurred in a triple junction formed by an isolated inclusion of cementite grain and two adjacent ferrite grains. The final microstructure is shown in Fig. 3.

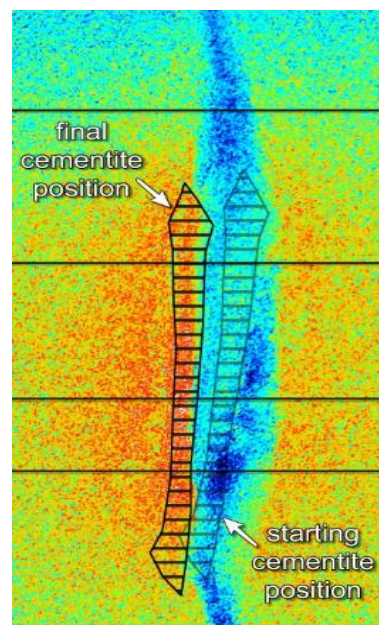


Fig. 3. Inspection of a steel sample where localized corrosion has occurred. Overlay of the original grain boundaries on the TEM image. The contours of the initial (gray) and final (black) place of cementite inclusion are presented [73]

Рис. 3. Обследование стального образца, где произошла локальная коррозия. Наложение исходных границ зерен на изображении ПЭМ. Представлены контуры начального (серого) и конечного (черного) места включения цементита [72]

The imposition of the initial contour of the grain boundary on the final image (Fig. 3) showed that the cementite grain shifted relative to its initial position.

The final photos showed that the ferrite was completely converted to an amorphous corrosion product after 1025 min of exposure. At this point, the ferrite regions of the sample lost about 30 nm, retaining 55 nm as iron oxide. Taking into account the initial thickness of 85 nm and the quantitative scale of the ferrite dissolution time, the rate of loss of corrosive material, due only to uniform corrosion, was calculated in the range from 0.015 to 0.16 mm per year at a penetration depth of 0.044 to 0.44 mm per year. It is expected that these values will be the upper limit, since they were calculated based on the possible contact of both surfaces of the sample with the solution.

Further understanding of the mechanisms of deposition of corrosion products can be achieved through the introduction of new tools and methods that complement TEM in situ and allow tracking the evolution of iron oxide in situ [74].

Nanoscale corrosion pathways like those identified in this article can have a profound impact on the degradation characteristics of materials. This work suggests that different types of electrochemically active processes will create percolation networks that corrode the inner body of steel much more quickly than predicted by uniform corrosion models.

#### CONCLUSION

High-strength materials, including carbon and low-alloy steels, as well as corrosion-resistant alloys, are necessary to overcome the material obstacles associated with the production of hydrocarbons from unconventional formations under high pressure and corrosive environments.

#### ЛИТЕРАТУРА

1. **Кудряшов В.С., Дерябкина А.В.** Анализ и перспективы освоения морских нефтегазовых ресурсов Арктики. *Экономика и управл. народ. хоз-вом.* 2020. № 13(5). С. 73-78.
2. **Бояринов А.Ю., Литвинова О.В.** Перспективы освоения Арктического шельфа. *Международ. науч.-иссл. журн.* 2021. № 2-2(104). С. 19-22. DOI: 10.23670/IRJ.2021.103.2.034.
3. **Бродт Л.Э.** Передовые практики нефтегазовых компаний по освоению газовых месторождений на Арктическом шельфе. *Арктика и Север.* 2021. № 44. С. 30-44. DOI: 10.37482/issn2221-2698.2021.44.30.
4. **Билинчук А.В., Ипатов А.И., Ситников А.В.** Промыслово-геофизический контроль разработки низкопроницаемых пластов в скважинах со сложным закачиванием. Опыт компании «Газпром нефть». *Нефт. хоз-во.* 2018. № 12. С. 34-37. DOI: 10.24887/0028-2448-2018-12-34-37.
5. **Колос В.Ю., Приходько А.С.** Моделирование гидросистем для контроля и регулирование процессов разработки месторождений. *Вестн. науки и образования.* 2019. № 11-3(65). С. 30-32.
6. **Guseynov Ch.S.** In Arctic - new technical means and technologies for the development of oil and gas fields in long-term freezing deepwater areas. *IOP Conf. Ser. Mater. Sci. Eng.* 2020. V. 734. Art. number 012174. DOI: 10.1088/1757-899X/734/1/012174.
7. **Конторович А.Э.** Пути освоения ресурсов нефти и газа Российского сектора Арктики. *Вестн. Российской акад. наук.* 2015. № 5-6(85). С. 420-430. DOI: 10.7868/S0869587315060171.
8. **Астафьев Д.А., Толстикова А.В., Наумова Л.А., Кабалин М.Ю.** Перспективные направления газонефтепоисковых работ на морском шельфе России в XXI веке. *Вестн. газовой науки.* 2018. № 4 (36). С. 4-18.
9. Offshore projects. Achievements and prospects [Electronic resource] URL: [https://shelf.gazprom-neft.ru/upload/iblock/248/2019\\_12\\_09\\_gpshsh\\_buklet\\_RUS.PDF](https://shelf.gazprom-neft.ru/upload/iblock/248/2019_12_09_gpshsh_buklet_RUS.PDF).
10. Official website of Shelf. Gazprom-neft [Electronic resource] URL: <https://prirazlomnoye.gazprom-neft.ru/development/social-responsibility-and-charity/>.
11. Official website of Rosneft. Business. Offshore production [Electronic resource] URL: <https://www.rosneft.ru/business/Upstream/offshore/>.

Environmental cracking and localized corrosion are two major forms of degradation that affect alloys and prevent them from operating safely and economically in high pressure, harsh marine environments and arctic fields. A better understanding of metallurgical factors and manufacturing variables that lead to optimal reduction in stress corrosion cracking is of paramount importance.

*The authors declare the absence a conflict of interest warranting disclosure in this article.*

*Авторы заявляют об отсутствии конфликта интересов, требующего раскрытия в данной статье.*

#### REFERENCES

1. **Kudryashov V.S., Deryabkina A.V.** Analysis and prospects for the development of offshore oil and gas resources of the Arctic. *Ekonom. Upravl. Narod. Khoz-vom.* 2020. N 13(5). P. 73-78 (in Russian).
2. **Boyarinov A.Yu., Litvinova O.V.** Prospects for the development of the Arctic shelf. *Mezhdunar. Nauch.-Issl. Zhurn.* 2021. N 2-2(104). P. 19-22 (in Russian). DOI: 10.23670/IRJ.2021.103.2.034.
3. **Brodth L.E.** Best practices of oil and gas companies in the development of gas fields on the Arctic shelf. *Arktika Sever.* 2021. N 44. P. 30-44 (in Russian). DOI: 10.37482/issn2221-2698.2021.44.30.
4. **Bilinchuk A.V., Ipatov A.I., Sitnikov A.V.** Field-geophysical control of the development of low-permeable formations in wells with complex injection. The experience of Gazprom Neft. *Neft. Khoz-vo.* 2018. N 12. P. 34-37 (in Russian). DOI: 10.24887/0028-2448-2018-12-34-37.
5. **Kolos V.Yu., Prihod'ko A.S.** Modeling of hydraulic systems for the control and regulation of field development processes. *Vestn. Nauki Obrazovaniya.* 2019. N 11-3(65). P. 30-32 (in Russian).
6. **Guseynov Ch.S.** In Arctic - new technical means and technologies for the development of oil and gas fields in long-term freezing deepwater areas. *IOP Conf. Ser. Mater. Sci. Eng.* 2020. V. 734. Art. number 012174. DOI: 10.1088/1757-899X/734/1/012174.
7. **Kontorovich A.E.** Ways to develop oil and gas resources of the Russian Arctic sector. *Vestn. Ross. Akad. Nauk.* 2015. N 5-6(85). P. 420-430 (in Russian). DOI: 10.7868/S0869587315060171.
8. **Astaf'ev D.A., Tolstikov A.V., Naumova L.A., Kabalin M.Yu.** Promising directions of gas and oil exploration work on the Russian offshore in the XXI century. *Vesti Gazovoy Nauki.* 2018. N 4 (36). P. 4-18 (in Russian).
9. Offshore projects. Achievements and prospects [Electronic resource] URL: [https://shelf.gazprom-neft.ru/upload/iblock/248/2019\\_12\\_09\\_gpshsh\\_buklet\\_RUS.PDF](https://shelf.gazprom-neft.ru/upload/iblock/248/2019_12_09_gpshsh_buklet_RUS.PDF).
10. Official website of Shelf. Gazprom-neft [Electronic resource] URL: <https://prirazlomnoye.gazprom-neft.ru/development/social-responsibility-and-charity/>.
11. Official website of Rosneft. Business. Offshore production [Electronic resource] URL: <https://www.rosneft.ru/business/Upstream/offshore/>.

12. Трошин М.С. Влияние международных экономических санкций на развитие экономики РФ. *Моск. эконом. журн.* 2021. № 3. С. 169-175. DOI: 10.24411/2413-046X-2021-10133.
13. Вагапов Р.К., Запевалов Д.Н., Ибатуллин К.А. Исследование коррозии объектов инфраструктуры газодобычи в присутствии CO<sub>2</sub> аналитическими методами контроля. *Вести газовой науки.* 2020. № 3(45). С. 81-92.
14. Прищепо Д., Хрулева Е., Пономарев А., Мартюшев Д., Сидоров М. Развитие отечественных технологий в области эксплуатации морских скважин арктического шельфа России. *Территория Нефтегаз.* 2019. № 3. С. 56-61.
15. Iannuzzi M., Barnoush A., Johnsen R. Materials and corrosion trends in offshore and subsea oil and gas production. *Mater. Degrad.* 2017. V. 1. N 1. P. 2. DOI: 10.1038/s41529-017-0003-4.
16. Лалетина А.С. ОПЕК. Газ. Трубопроводы. Право. М.: Проспект. 2019. 128 с.
17. Hou B., Li X., Ma X., Du C., Zhang D., Zheng M., Xu W., Lu D., Ma F. The cost of corrosion in China. *Mater. Degrad.* 2017. V. 1. N 1. P. 4. DOI: 10.1038/s41529-017-0005-2.
18. Tawancy H.M., Al-Hadhrani L.M., Al-Yousef F.K. Analysis of corroded elbow section of carbon steel piping system of an oil-gas separator vessel. *Case Stud. Eng. Fail. Anal.* 2013. V. 1. N 1. P. 6-14. DOI: 10.1016/j.csefa.2012.11.001.
19. Cheng Y.F. Stress corrosion cracking of pipelines. In: Wiley Series in Corrosion. Ed. by R.V. Revie. USA: John Wiley & Sons. 2013. V. 15. P. 4-10. DOI: 10.1002/9781118537022.
20. Vanaei H.R., Eslami A., Egbewande A. A review on pipeline corrosion, in-line inspection (ILI), and corrosion growth rate models. *Internat. J. Pres. Vessels Piping.* 2017. V. 149. P. 43-54. DOI: 10.1016/j.ijpvp.2016.11.007.
21. Матризаев М.Ю., Халлыев Н.Х. Особенности проектирования морских промысловых трубопроводов в современных условиях. *Petrol. Eng.* 2019. № 2(17). С. 104-110. DOI: 10.17122/ngdelo-2019-2-104-110.
22. Chen M., Zhang H., Chen L., Fu D. An electrochemical method based on OCP fluctuations for anti-corrosion alloy composition optimization. *Anti-Corros. Methods Mater.* 2018. V. 65. N 3. P. 325-330. DOI: 10.1108/ACMM-03-2018-1913.
23. Bruschi R., Bartolini L. M., Cherubini P., Torselletti E., Vitali L. Meeting the challenges of the upcoming development of subsea fields: from carbon steel pipelines to new materials and pipe concepts. Proc. of the 27th Internat. Ocean and Polar Eng. Conf. San Francisco, California. 2017. P. 1-21.
24. Warke W.R. Stress-Corrosion Cracking. In: Failure Analysis and Prevention. Ed. by W.T. Becker, R.J. Shipley. ASM International. 2002. V. 11. DOI: 10.31399/asm.hb.v11.9781627081801.
25. Кушнаренко В.М., Чирков Ю.А., Пояркова Е.В., Клещарева Г.А. Оценка сопротивления сталей сероводородному растрескиванию. Ст. в Сб. тр. конф. «Передовые научно-технические и социально-гуманитарные проекты в современной науке». 2019. С. 21.
26. Сагадеев И.Р., Ерофеев В.В., Шарафиев Р.Г., Альмухаметов А.А., Киреев И.Р., Гильманшин Р.А., Якупов В.М. Исследование механизма коррозионного разрушения металла в сероводородсодержащих средах. *Нефтегаз. дело.* 2017. № 1(15). С. 99-102.
27. Давыдкин М.В., Немыкина О.В. Испытания сталей на сероводородное коррозионное растрескивание. *Коррозия: материалы и защита.* 2015. № 11. С. 18-25.
12. Troshin M.S. The impact of international economic sanctions on the development of the Russian economy. *Mosk. Ekonom. Zhurn.* 2021. N 3. P. 169-175 (in Russian). DOI: 10.24411/2413-046X-2021-10133.
13. Vagapov R.K., Zapevalov D.N., Ibatullin K.A. Investigation of corrosion of gas production infrastructure facilities in the presence of CO<sub>2</sub> by analytical control methods. *Vesti Gazovoy Nauki.* 2020. N 3(45). P. 81-92 (in Russian).
14. Prishchepo D., Khruleva E., Ponomarev A., Martyshev D., Sidorov M. Development of domestic technologies in the field of exploitation of offshore wells of the Arctic shelf of Russia. *Territoriya Neftegaz.* 2019. N 3. P. 56-61 (in Russian).
15. Iannuzzi M., Barnoush A., Johnsen R. Materials and corrosion trends in offshore and subsea oil and gas production. *Mater. Degrad.* 2017. V. 1. N 1. P. 2. DOI: 10.1038/s41529-017-0003-4.
16. Laletina A.S. OPEC. Gaz. Truboprovody. Pravo. M.: Prospekt. 2019. 128 p. (in Russian).
17. Hou B., Li X., Ma X., Du C., Zhang D., Zheng M., Xu W., Lu D., Ma F. The cost of corrosion in China. *Mater. Degrad.* 2017. V. 1. N 1. P. 4. DOI: 10.1038/s41529-017-0005-2.
18. Tawancy H.M., Al-Hadhrani L.M., Al-Yousef F.K. Analysis of corroded elbow section of carbon steel piping system of an oil-gas separator vessel. *Case Stud. Eng. Fail. Anal.* 2013. V. 1. N 1. P. 6-14. DOI: 10.1016/j.csefa.2012.11.001.
19. Cheng Y.F. Stress corrosion cracking of pipelines. In: Wiley Series in Corrosion. Ed. by R.V. Revie. USA: John Wiley & Sons. 2013. V. 15. P. 4-10. DOI: 10.1002/9781118537022.
20. Vanaei H.R., Eslami A., Egbewande A. A review on pipeline corrosion, in-line inspection (ILI), and corrosion growth rate models. *Internat. J. Pres. Vessels Piping.* 2017. V. 149. P. 43-54. DOI: 10.1016/j.ijpvp.2016.11.007.
21. Matrizayev M.Yu., Khallyev N.H. Design features of offshore field pipelines in modern conditions. *Petrol. Eng.* 2019. N 2(17). P. 104-110 (in Russian). DOI: 10.17122/ngdelo-2019-2-104-110.
22. Chen M., Zhang H., Chen L., Fu D. An electrochemical method based on OCP fluctuations for anti-corrosion alloy composition optimization. *Anti-Corros. Methods Mater.* 2018. V. 65. N 3. P. 325-330. DOI: 10.1108/ACMM-03-2018-1913.
23. Bruschi R., Bartolini L. M., Cherubini P., Torselletti E., Vitali L. Meeting the challenges of the upcoming development of subsea fields: from carbon steel pipelines to new materials and pipe concepts. Proc. of the 27th Internat. Ocean and Polar Eng. Conf. San Francisco, California. 2017. P. 1-21.
24. Warke W.R. Stress-Corrosion Cracking. In: Failure Analysis and Prevention. Ed. by W.T. Becker, R.J. Shipley. ASM International. 2002. V. 11. DOI: 10.31399/asm.hb.v11.9781627081801.
25. Kushnarenko V.M., Chirkov Yu.A., Poyarkova E.V., Kleshchareva G.A. Evaluation of the resistance of steels to hydrogen sulfide cracking. Art. in coll. of conf. mater. «Advanced scientific-technical and socio-humanitarian projects in modern science». 2019. P. 21 (in Russian).
26. Sagadeev I.R., Erofeev V.V., Sharafiev R.G., Al'mukhametov A.A., Kireev I.R., Gil'manshin R.A., Yakupov V.M. Investigation of the mechanism of corrosion destruction of metal in hydrogen sulfide-containing media. *Neftegaz. Delo.* 2017. N 1(15). P. 99-102 (in Russian).
27. Davydkin M.V., Nemykina O.V. Testing of steels for hydrogen sulfide corrosion cracking. *Korroziya: Mater. Zashchita.* 2015. N 11. P. 18-25 (in Russian).
28. Bulkeshev D.O. Investigation of stress intensity in the process of corrosion cracking of steel of main gas pipelines. *Universum: Tekhn. Nauki.* 2019. N 2(59). P. 17-21 (in Russian).



28. Булклешев Д.О. Исследование интенсивности напряжений в процессе коррозионного растрескивания стали магистральных газопроводов. *Universum: Техн. науки*. 2019. № 2(59). С. 17-21
29. Сергеев Н.Н., Извольский В.В., Сергеев А.Н., Кутепов С.Н. Влияние химического состава стали 23x2г2т на стойкость против коррозионного растрескивания. *Изв. Тул. гос. ун-та*. 2019. № 9. С. 409-420.
30. Степанов П.П., Барыков А.М., Парунов А.Б., Елагина О.Ю. Стойкость металла труб из высокопрочных сталей к водородному растрескиванию в морских средах, содержащих сероводород. *Наука и техника в газ. пром-сти*. 2015. № 1. С. 51-56.
31. Pioszak G.L., Gangloff R.P. Hydrogen environment assisted cracking of modern ultra-high strength martensitic steels. *Metalurg. Mater. Transact. A*. 2017. V. 48. N 9. P. 4025-4045. DOI: 10.1007/s11661-017-4156-0.
32. Nešić S. Key issues related to modelling of internal corrosion of oil and gas pipelines – A review. *Corrosion Sci.* 2007. V. 49. N 12. P. 4308-4338. DOI: 10.1016/j.corsci.2007.06.006.
33. Крылов П.В., Ширяев А.Г., Чикалов С.Г., Пышминцев И.Ю., Четвериков С.Г., Рескин С.А. Разработка марок стали для труб повышенной и высокой прочности, стойких к углекислотной коррозии, в хладостойком исполнении. *Территория «NEFTEGAS»*. 2017. № 12. С. 48-53.
34. Чечурнина М.Н., Соколенко В.Э. Сравнительный анализ национальных инновационных систем стран Арктического региона. *Экономика: вчера, сегодня и завтра*. 2017. № 9А (7). С. 115-131.
35. Муленко В.В., Сапрыкина К.М. Экологические и экономические риски разработки морских нефтегазовых месторождений Крайнего Севера. *Территория «NEFTEGAS»*. 2016. № 2. С. 94-99.
36. Alvaro A., Akselsen O. M., Ren X., Kane A. in Proc. of the Twenty-fourth Internat. Ocean and Polar Eng. Conf. 2014. P. 247–254.
37. Oryshchenko A.S., Malyshevskii V.A., Petrov S.N., Shumilov E.A. Relation between the Degree of Alloying, Structure, And Mechanical Properties of High-Strength Steel. *Steel Translation*. 2018. V. 48. N 3. P. 143-148. DOI: 10.3103/S0967091218030099.
38. Craig B.D. On the contradiction of applying rolled threads to bolting exposed to hydrogen-bearing environments. *Oil Gas Facilities*. 2015. V. 4. N 06. P. 66-71. DOI: 10.2118/178431-PA.
39. Долгих С.А., Ибрагимов Н.Г., Ткачева В.Э., Шакиров Ф.Ш. Опыт эксплуатации системы катодной защиты обсадных колонн скважин на территории ПАО «ТАТНЕФЕТЬ». Ст. в сб. науч. тр. «ТАТНИПИНЕФТЬ». 2017. С. 467-473.
40. Выбойщик М.А., Иоффе А.В. Разработка стали к углекислотной коррозии в нефтедобываемых средах. *Перспектив. матер.* 2017. С. 115-166.
41. Зырянов А.О., Иоффе А.В., Тетюева Т.В., Выбойщик М.А. Особенности коррозии НКТ из стали 15Х5МФБЧ в средах, содержащих CO<sub>2</sub> и CO<sub>2</sub>/H<sub>2</sub>S. Ст. в сб. тр. конф. «Прочность неоднородных структур – Прост 2018». 2018. С. 198.
42. Kane R.D., Wilhelm S.M., Oldfield J.W. Review of Hydrogen Induced Cracking of Steels in Wet H<sub>2</sub>S Refinery Service. In: International Conference on Interaction of Steels with Hydrogen in Petroleum Industry Pressure Vessel Service. New York, NY: Materials Properties Council. 1989. 28 March, Paris, France.
43. Kappes M., Brownlee K., Bruno T.V. Sulfide stress cracking of nickel-containing low-alloy steels. *Corros. Rev.* 2014. V. 32. N 3-4. P. 101-128. DOI: 10.1515/correv-2014-0027.
44. Snape E. Roles of composition and microstructure in sulfide cracking of steel. *Corrosion*. 1968. V. 24. N 9. P. 261-282. DOI: 10.5006/0010-9312-24.9.261.
29. Sergeev N.N., Izvol'skiy V.V., Sergeev A.N., Kutepov S.N. The influence of the chemical composition of steel 23x2g2t on the resistance against corrosion cracking. *Izv. Tul. Gos. Univ.* 2019. N 9. P. 409-420 (in Russian).
30. Stepanov P.P., Barykov A.M., Parunov A.B., Elagina O.Yu. Resistance of metal pipes made of high-strength steels to hydrogen cracking in marine environments containing hydrogen sulfide. *Nauka Tekhnika Gaz. Prom-sti*. 2015. N 1. P. 51-56 (in Russian).
31. Pioszak G.L., Gangloff R.P. Hydrogen environment assisted cracking of modern ultra-high strength martensitic steels. *Metalurg. Mater. Transact. A*. 2017. V. 48. N 9. P. 4025-4045. DOI: 10.1007/s11661-017-4156-0.
32. Nešić S. Key issues related to modelling of internal corrosion of oil and gas pipelines – A review. *Corrosion Sci.* 2007. V. 49. N 12. P. 4308-4338. DOI: 10.1016/j.corsci.2007.06.006.
33. Krylov P.V., Shiryaev A.G., Chikalov S.G., Pyshmintsev I.Yu., Chetverikov S.G., Reskin S.A. Development of steel grades for pipes of increased and high strength, resistant to carbon dioxide corrosion, in cold-resistant design. *Territoriya «NEFTEGAS»*. 2017. N 12. P. 48-53 (in Russian).
34. Chechurina M.N., Sokolenko V.E. Comparative analysis of national innovation systems of the countries of the Arctic region. *Ekonomika: Vchera, Segodnya Zavtra*. 2017. N 9A (7). P. 115-131 (in Russian).
35. Mulenko V.V., Saprykina K.M. Environmental and economic risks of developing offshore oil and gas fields in the Far North. *Territoriya «NEFTEGAS»*. 2016. N 2. P. 94-99 (in Russian).
36. Alvaro A., Akselsen O. M., Ren X., Kane A. in Proc. of the Twenty-fourth Internat. Ocean and Polar Eng. Conf. 2014. P. 247–254.
37. Oryshchenko A.S., Malyshevskii V.A., Petrov S.N., Shumilov E.A. Relation between the Degree of Alloying, Structure, And Mechanical Properties of High-Strength Steel. *Steel Translation*. 2018. V. 48. N 3. P. 143-148. DOI: 10.3103/S0967091218030099.
38. Craig B.D. On the contradiction of applying rolled threads to bolting exposed to hydrogen-bearing environments. *Oil Gas Facilities*. 2015. V. 4. N 06. P. 66-71. DOI: 10.2118/178431-PA.
39. Dolgikh S.A., Ibragimov N.G., Tkacheva V.E., Shakirov F.Sh. Experience in the operation of the cathodic protection system of well casings on the territory of PJSC TATNEFET. Art. in coll. of sci. mater. «TATNIPINEFT». 2017. P. 467-473 (in Russian).
40. Vyboyschchik M.A., Ioffe A.V. Development of steel for carbon dioxide corrosion in oil-producing environments. *Perspektiv. mater.* 2017. P. 115-166 (in Russian).
41. Zyryanov A.O., Ioffe A.V., Tetyueva T.V., Vyboyschchik M.A. Corrosion features of tubing made of 15X5MFBH steel in environments containing CO<sub>2</sub> and CO<sub>2</sub>/H<sub>2</sub>S. Art. in coll. of conf. mater. «Prochnost' neodnorodnyh struktur – Prost 2018». 2018. P. 198 (in Russian).
42. Kane R.D., Wilhelm S.M., Oldfield J.W. Review of Hydrogen Induced Cracking of Steels in Wet H<sub>2</sub>S Refinery Service. In: International Conference on Interaction of Steels with Hydrogen in Petroleum Industry Pressure Vessel Service. New York, NY: Materials Properties Council. 1989. 28 March, Paris, France.
43. Kappes M., Brownlee K., Bruno T.V. Sulfide stress cracking of nickel-containing low-alloy steels. *Corros. Rev.* 2014. V. 32. N 3-4. P. 101-128. DOI: 10.1515/correv-2014-0027.
44. Snape E. Roles of composition and microstructure in sulfide cracking of steel. *Corrosion*. 1968. V. 24. N 9. P. 261-282. DOI: 10.5006/0010-9312-24.9.261.

43. **Kappes M., Brownlee K., Bruno T.V.** Sulfide stress cracking of nickel-containing low-alloy steels. *Corros. Rev.* 2014. V. 32. N 3-4. P. 101-128. DOI: 10.1515/corrrev-2014-0027.
44. **Snape E.** Roles of composition and microstructure in sulfide cracking of steel. *Corrosion.* 1968. V. 24. N 9. P. 261-282. DOI: 10.5006/0010-9312-24.9.261.
45. **Fielding L.C.D.** The bainite controversy. *Mater. Sci. Technol.* 2013. V. 29. N 4. P. 383-399. DOI: 10.1179/1743284712Y.0000000157.
46. **Cancio M., Eloff B., Kissner G., Valdez M., Vouilloz F.J.** High strength low alloy steel for HPHT wells. In: Proc. of the Annual Offshore Technology Conf. 2014. P. 1. DOI: 10.2118/24746-MS.
47. **Randle V.** Grain boundary engineering: an overview after 25 years. *Mater. Sci. Technol.* 2010. V. 26. N 3. P. 253-261. DOI: 10.1179/026708309X12601952777747.
48. **Kuzmina M., Ponge D., Raabe D.** Grain boundary segregation engineering and austenite reversion turn embrittlement into toughness: example of a 9 wt.% medium Mn steel. *Acta Materialia.* 2015. V. 86. P. 182-192. DOI: 10.1016/j.actamat.2014.12.021.
49. **Watanabe T.** Grain boundary engineering: historical perspective and future prospects. *J. Mater. Sci.* 2011. V. 46. N 12. P. 4095-4115. DOI: 10.1007/s10853-011-5393-z.
50. **Rhodes P.R., Skogsberg L.A., Tuttle R.N.** Pushing the limits of metals in corrosive oil and gas well environments. *Corrosion.* 2007. V. 63. N 1. P. 63-100. DOI: 10.5006/1.3278334.
51. **Talbot D.E.J., Talbot J.D.R.** Corrosion science and technology. CRC press. 2018. 596 p.
52. **Bruemmer S.M., Olszta M.J., Toloczko M.B., Schreiber D.K.** Grain boundary selective oxidation and intergranular stress corrosion crack growth of high-purity nickel binary alloys in high-temperature hydrogenated water. *Corros. Sci.* 2018. V. 131. P. 310-323. DOI: 10.1016/j.corsci.2017.11.024.
53. **Xie Y., Wu Y., Burns J., Zhang J.** Characterization of stress corrosion cracks in Ni-based weld alloys 52, 52M and 152 grown in high-temperature water. *Mater. Character.* 2016. V. 112. P. 87-97. DOI: 10.1016/j.matchar.2015.12.005.
54. **Rebak R.B., Ortiz M.R., Iannuzzi M., Kappes M., Mishra A., Rodriguez M.A.** Effect of thermal treatment on the localized corrosion behavior of alloy 718. In: EUROCORR. 2014. P. 8-12.
55. **Dong J.X., Zhang M.C., Mannan S.K.** Microstructures and the structure stability of inconel 725, a new age - hardenable corrosion resistant superalloy. *Acta Metallurgica Sinica.* 2009. V. 16. N 2. P. 145-150.
56. **Yoo S.C., Choi K.J., Kim T., Kim S.H., Kim J.Yo., Kim J.H.** Microstructural evolution and stress-corrosion-cracking behavior of thermally aged Ni-Cr-Fe alloy. *Corros. Sci.* 2016. V. 111. P. 39-51. DOI: 10.1016/j.corsci.2016.04.051.
57. **You X., Tan Y., Zhao L., You Q., Wang Y., Ye F., Li J.** Effect of solution heat treatment on microstructure and electrochemical behavior of electron beam smelted Inconel 718 superalloy. *J. Alloys Comp.* 2018. V. 741. P. 792-803. DOI: 10.1016/j.jallcom.2018.01.159.
58. **Paxton A.T., Katzarov I.H.** Quantum and isotope effects on hydrogen diffusion, trapping and escape in iron. *Acta Materialia.* 2016. V. 103. P. 71-76. DOI: 10.1016/j.actamat.2015.09.054.
59. **Yakubov V., Lin M., Volinsky A.A., Qiao L., Guo L.** The hydrogen-induced pitting corrosion mechanism in duplex stainless steel studied by current-sensing atomic force microscopy. *Mater. Degrad.* 2018. V. 2. Art. number 39. DOI: 10.1038/s41529-018-0062-1.
60. **Thomas S., Sundararajan G., White P., Birbilis N.** The effect of absorbed hydrogen on the corrosion of steels: review, discussion, and implications. *Corrosion.* 2017. V. 73. N 4. P. 426-436. DOI: 10.5006/2242.
45. **Fielding L.C.D.** The bainite controversy. *Mater. Sci. Technol.* 2013. V. 29. N 4. P. 383-399. DOI: 10.1179/1743284712Y.0000000157.
46. **Cancio M., Eloff B., Kissner G., Valdez M., Vouilloz F.J.** High strength low alloy steel for HPHT wells. In: Proc. of the Annual Offshore Technology Conf. 2014. P. 1. DOI: 10.2118/24746-MS.
47. **Randle V.** Grain boundary engineering: an overview after 25 years. *Mater. Sci. Technol.* 2010. V. 26. N 3. P. 253-261. DOI: 10.1179/026708309X12601952777747.
48. **Kuzmina M., Ponge D., Raabe D.** Grain boundary segregation engineering and austenite reversion turn embrittlement into toughness: example of a 9 wt.% medium Mn steel. *Acta Materialia.* 2015. V. 86. P. 182-192. DOI: 10.1016/j.actamat.2014.12.021.
49. **Watanabe T.** Grain boundary engineering: historical perspective and future prospects. *J. Mater. Sci.* 2011. V. 46. N 12. P. 4095-4115. DOI: 10.1007/s10853-011-5393-z.
50. **Rhodes P.R., Skogsberg L.A., Tuttle R.N.** Pushing the limits of metals in corrosive oil and gas well environments. *Corrosion.* 2007. V. 63. N 1. P. 63-100. DOI: 10.5006/1.3278334.
51. **Talbot D.E.J., Talbot J.D.R.** Corrosion science and technology. CRC press. 2018. 596 p.
52. **Bruemmer S.M., Olszta M.J., Toloczko M.B., Schreiber D.K.** Grain boundary selective oxidation and intergranular stress corrosion crack growth of high-purity nickel binary alloys in high-temperature hydrogenated water. *Corros. Sci.* 2018. V. 131. P. 310-323. DOI: 10.1016/j.corsci.2017.11.024.
53. **Xie Y., Wu Y., Burns J., Zhang J.** Characterization of stress corrosion cracks in Ni-based weld alloys 52, 52M and 152 grown in high-temperature water. *Mater. Character.* 2016. V. 112. P. 87-97. DOI: 10.1016/j.matchar.2015.12.005.
54. **Rebak R.B., Ortiz M.R., Iannuzzi M., Kappes M., Mishra A., Rodriguez M.A.** Effect of thermal treatment on the localized corrosion behavior of alloy 718. In: EUROCORR. 2014. P. 8-12.
55. **Dong J.X., Zhang M.C., Mannan S.K.** Microstructures and the structure stability of inconel 725, a new age - hardenable corrosion resistant superalloy. *Acta Metallurgica Sinica.* 2009. V. 16. N 2. P. 145-150.
56. **Yoo S.C., Choi K.J., Kim T., Kim S.H., Kim J.Yo., Kim J.H.** Microstructural evolution and stress-corrosion-cracking behavior of thermally aged Ni-Cr-Fe alloy. *Corros. Sci.* 2016. V. 111. P. 39-51. DOI: 10.1016/j.corsci.2016.04.051.
57. **You X., Tan Y., Zhao L., You Q., Wang Y., Ye F., Li J.** Effect of solution heat treatment on microstructure and electrochemical behavior of electron beam smelted Inconel 718 superalloy. *J. Alloys Comp.* 2018. V. 741. P. 792-803. DOI: 10.1016/j.jallcom.2018.01.159.
58. **Paxton A.T., Katzarov I.H.** Quantum and isotope effects on hydrogen diffusion, trapping and escape in iron. *Acta Materialia.* 2016. V. 103. P. 71-76. DOI: 10.1016/j.actamat.2015.09.054.
59. **Yakubov V., Lin M., Volinsky A.A., Qiao L., Guo L.** The hydrogen-induced pitting corrosion mechanism in duplex stainless steel studied by current-sensing atomic force microscopy. *Mater. Degrad.* 2018. V. 2. Art. number 39. DOI: 10.1038/s41529-018-0062-1.
60. **Thomas S., Sundararajan G., White P., Birbilis N.** The effect of absorbed hydrogen on the corrosion of steels: review, discussion, and implications. *Corrosion.* 2017. V. 73. N 4. P. 426-436. DOI: 10.5006/2242.
61. **Wang X.Z., Luo H., Luo J.L.** Effects of hydrogen and stress on the electrochemical and passivation behaviour of 304 stainless steel in simulated PEMFC environment. *Electrochim. Acta.* 2019. V. 293. P. 60-77. DOI: 10.1016/j.electacta.2018.10.028.

61. Wang X.Z., Luo H., Luo J.L. Effects of hydrogen and stress on the electrochemical and passivation behaviour of 304 stainless steel in simulated PEMFC environment. *Electrochim. Acta*. 2019. V. 293. P. 60-77. DOI: 10.1016/j.electacta.2018.10.028.
62. Deng Y., Hajilou T., Wan D., Kheradmand N., Barnoush A. In-situ micro-cantilever bending test in environmental scanning electron microscope: Real time observation of hydrogen enhanced cracking. *Scripta Materialia*. 2017. V. 127. P. 19-23. DOI: 10.1016/j.scriptamat.2016.08.026.
63. Hajilou T., Deng Y., Rogne B. R., Kheradmand N., Barnoush A. In situ electrochemical microcantilever bending test: A new insight into hydrogen enhanced cracking. *Scripta Materialia*. 2017. V. 132. P. 17-21. DOI: 10.1016/j.scriptamat.2017.01.019.
64. Golovin Y.I., Korenkov V.V., Razlivalova S.S. The Effect of Low-Amplitude Oscillations of the Load on the Local Stiffness of Materials in Loaded Nanocontacts Detected by Methods of Dynamic Nanoindentation. *Adv. Mater. Technol.* 2018. V. 2. P. 3-8. DOI: 10.17277/amt.2018.02.pp.003-008.
65. Wu D., Jang J.S.C., Nieh T.G. Elastic and plastic deformations in a high entropy alloy investigated using a nanoindentation method. *Intermetallics*. 2016. V. 68. P. 118-127. DOI: 10.1016/j.intermet.2015.10.002.
66. Lai H.D., Jian S.-R., Tuyen L.T.C., Le P.H., Luo C.-W., Juang J.-Y. Nanoindentation of Bi<sub>2</sub>Se<sub>3</sub> thin films. *Micromachines*. 2018. V. 9. N 10. P. 518. DOI: 10.3390/mi9100518.
67. Martin M. L., Pundt A., Kirchheim R. Hydrogen-induced accelerated grain growth in vanadium. *Acta Materialia*. 2018. V. 155. P. 262-267. DOI: 10.1016/j.actamat.2018.06.011.
68. Sun J.B., Zhang G.A., Liu W., Lu M.X. The formation mechanism of corrosion scale and electrochemical characteristic of low alloy steel in carbon dioxide-saturated solution. *Corrosion Sci.* 2012. V. 57. P. 131-138. DOI: 10.1016/j.corsci.2011.12.025.
69. Wranglen G. Pitting and sulphide inclusions in steel. *Corrosion Sci.* 1974. V. 14. P. 331-349. DOI: 10.1016/S0010-938X(74)80047-8.
70. Bertali G., Scenini F., Burke M. The effect of residual stress on the preferential intergranular oxidation of alloy 600. *Corrosion Sci.* 2016. V. 111. P. 494-507. DOI: 10.1016/j.corsci.2016.05.022.
71. Нгуен В.Ч., Као Н.Л., Донг В.К. Исследование коррозии конструкционной стали АН-36 в морской среде Вьетнама. *Изв. вузов. Химия и хим. технология*. 2021. Т. 64. Вып. 10. С. 139-144. DOI: 10.6060/ivkkt.20216410.6496.
72. Carter C.B., Williams D.B. Transmission electron microscopy: Diffraction, imaging, and spectrometry. Springer. 2016. 543 p.
73. Hayden S.C., Chisholm C., Grudt R.O., Aguiar J.A., Mook W.M., Kotula P.G., Pilyugina T.S., Bufford D.C., Hattar Kh., Kucharski T.J., Taie I.M., Ostraat M.L., Jungjohann K.L. Localized corrosion of low-carbon steel at the nanoscale. *Mater. Degrad.* 2019. V. 3. N 1. P. 17. DOI: 10.1038/s41529-019-0078-1.
74. Jungjohann K.L., Evans J.E., Aguiar J.A., Arslan I., Browning N.D. Atomic-scale imaging and spectroscopy for in situ liquid scanning transmission electron microscopy. *Microsc. Microanal.* 2012. V. 18. P. 621-627. DOI: 10.1017/S1431927612000104.
62. Deng Y., Hajilou T., Wan D., Kheradmand N., Barnoush A. In-situ micro-cantilever bending test in environmental scanning electron microscope: Real time observation of hydrogen enhanced cracking. *Scripta Materialia*. 2017. V. 127. P. 19-23. DOI: 10.1016/j.scriptamat.2016.08.026.
63. Hajilou T., Deng Y., Rogne B. R., Kheradmand N., Barnoush A. In situ electrochemical microcantilever bending test: A new insight into hydrogen enhanced cracking. *Scripta Materialia*. 2017. V. 132. P. 17-21. DOI: 10.1016/j.scriptamat.2017.01.019.
64. Golovin Y.I., Korenkov V.V., Razlivalova S.S. The Effect of Low-Amplitude Oscillations of the Load on the Local Stiffness of Materials in Loaded Nanocontacts Detected by Methods of Dynamic Nanoindentation. *Adv. Mater. Technol.* 2018. V. 2. P. 3-8. DOI: 10.17277/amt.2018.02.pp.003-008.
65. Wu D., Jang J.S.C., Nieh T.G. Elastic and plastic deformations in a high entropy alloy investigated using a nanoindentation method. *Intermetallics*. 2016. V. 68. P. 118-127. DOI: 10.1016/j.intermet.2015.10.002.
66. Lai H.D., Jian S.-R., Tuyen L.T.C., Le P.H., Luo C.-W., Juang J.-Y. Nanoindentation of Bi<sub>2</sub>Se<sub>3</sub> thin films. *Micromachines*. 2018. V. 9. N 10. P. 518. DOI: 10.3390/mi9100518.
67. Martin M. L., Pundt A., Kirchheim R. Hydrogen-induced accelerated grain growth in vanadium. *Acta Materialia*. 2018. V. 155. P. 262-267. DOI: 10.1016/j.actamat.2018.06.011.
68. Sun J.B., Zhang G.A., Liu W., Lu M.X. The formation mechanism of corrosion scale and electrochemical characteristic of low alloy steel in carbon dioxide-saturated solution. *Corrosion Sci.* 2012. V. 57. P. 131-138. DOI: 10.1016/j.corsci.2011.12.025.
69. Wranglen G. Pitting and sulphide inclusions in steel. *Corrosion Sci.* 1974. V. 14. P. 331-349. DOI: 10.1016/S0010-938X(74)80047-8.
70. Bertali G., Scenini F., Burke M. The effect of residual stress on the preferential intergranular oxidation of alloy 600. *Corrosion Sci.* 2016. V. 111. P. 494-507. DOI: 10.1016/j.corsci.2016.05.022.
71. Nguen V.Ch., Kao N.L., Dong V.K. Investigation of corrosion of AN-36 structural steel in the marine environment of Vietnam. *ChemChemTech [Izv. Vyssh. Uchebn. Zaved. Khim. Khim. Tekhnol.]* 2021. V. 64. N 10. P. 139-144 (in Russian). DOI: 10.6060/ivkkt.20216410.6496.
72. Carter C.B., Williams D.B. Transmission electron microscopy: Diffraction, imaging, and spectrometry. Springer. 2016. 543 p.
73. Hayden S.C., Chisholm C., Grudt R.O., Aguiar J.A., Mook W.M., Kotula P.G., Pilyugina T.S., Bufford D.C., Hattar Kh., Kucharski T.J., Taie I.M., Ostraat M.L., Jungjohann K.L. Localized corrosion of low-carbon steel at the nanoscale. *Mater. Degrad.* 2019. V. 3. N 1. P. 17. DOI: 10.1038/s41529-019-0078-1.
74. Jungjohann K.L., Evans J.E., Aguiar J.A., Arslan I., Browning N.D. Atomic-scale imaging and spectroscopy for in situ liquid scanning transmission electron microscopy. *Microsc. Microanal.* 2012. V. 18. P. 621-627. DOI: 10.1017/S1431927612000104.

Поступила в редакцию 03.10.2022

Принята к опубликованию 23.01.2023

Received 03.10.2022

Accepted 23.01.2023