

## РЕАКЦИОННАЯ СПОСОБНОСТЬ АЛЮМООКСИДНЫХ ПРЕКУРСОРОВ ДЛЯ ТВЕРДОФАЗНОГО ОБРАЗОВАНИЯ ШПИНЕЛИ $MgAl_2O_4$

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*Сопоставлена реакционная способность различных алюмооксидных прекурсоров в реакции образования магнезиальной шпинели: промышленного производства (корундового порошка КП (KP), глиноземов: металлургического ГК (GK), неметаллургического Г-00 (G-00), реактивного РГ (RG)) и продукта горения ксерогеля из нитрата алюминия с лимонной кислотой в условиях механоактивирующей обработки и без нее. Проанализированы ИК-спектры корунда и периклаза после механической обработки разного типа (ударно-стирающей в планетарной мельнице, истирания в шаро-кольцевой мельнице). Установлено, что кратковременное воздействие не только активировало компоненты, но и способствовало разрушению адсорбционных соединений на стадии подготовки. Проанализировано влияние механоактивации реагентов на выход шпинели. Установлено, что алюмооксидный продукт горения ксерогеля без отжига являлся рентгеноаморфным, что предполагало его высокую химическую активность. При его использовании обнаружено более полное связывание исходных реагентов в продукт по сравнению со смесью периклаза и корунда. Даже в отсутствие механоактивации компонентов удавалось достичь высокого выхода шпинели (до 80%). Полученные значения эффективных констант скорости реакции свидетельствуют о большей результативности механической обработки, включающей в себя ударную компоненту, при совместной активации периклаза и корунда/глинозема. Совместная обработка реагентов в планетарной мельнице (ПМ) позволила увеличить скорость реакции в 5-6,5 раз, тогда как истирание в шаро-кольцевой мельнице – всего в 1,7 раза. Предварительная обработка одного из компонентов исходной смеси в ПМ наиболее целесообразна для периклаза, поскольку она дала возможность ускорить реакцию в ~ 5 раз, а также наиболее выгодна энергетически и технологически из-за обработки только одного компонента пониженной твердости. Применение в синтезе шпинели продукта горения ксерогеля весьма эффективно, т.к. ускорило процесс в ~4 раза даже без предварительной активации механическим способом.*

**Ключевые слова:** шпинель,  $MgAl_2O_4$ , твердофазный синтез, кинетика, механоактивация, корунд, глинозем, периклаз, механоактивация, ИК-спектры, продукт горения ксерогеля

## REACTIVITY OF ALUMINUM OXIDE PRECURSORS FOR SOLID-PHASE FORMATION OF $MgAl_2O_4$ SPINEL

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*The reactivity of various alumina precursors in the reaction of magnesia spinel formation is compared: industrial powders (corundum powder KP, alumina: metallurgical GC, non-metallurgical G-00, reactive RG) and the xerogel combustion product from aluminum nitrate with citric acid under conditions of mechanical activating treatment and without it. The infrared spectra of corundum and periclase after mechanical treatment of various types (impact-abrasion in a planetary mill, abrasion in a ball-ring mill) were analyzed. It has been found that short-term action not only activated the components, but also contributed to the destruction of adsorption compounds at the preparation stage. The effect of mechanical activation of reagents on spinel yield was analyzed. It has been established that the alumina xerogel combustion product without annealing was X-ray amorphous, which suggested its high chemical activity. When using it, a more complete binding of initial reagents into the product was found compared to a mixture of periclase and corundum. Even in the absence of mechanical activation of the components, it was possible to achieve a high spinel yield (up to 80%). The obtained values of the effective reaction rate constants indicated a greater efficiency of mechanical treatment, including an impact component, with the combined activation of periclase and corundum/alumina. Co-treatment of reagents in a planetary mill (PM) made it possible to increase the reaction rate by 6-6.5 times, while abrasion in a ball-ring mill was only 1.7 times. Pretreatment of one component of the initial mix in PM was the most expedient for periclase, since it allowed to speed up the reaction by ~ 5 times, and is also the most profitable energetically and technologically due to the treatment of only one component of reduced hardness. The use of xerogel combustion product in the spinel synthesis was very effective, because it accelerated the process by ~4 times even without preliminary activation by mechanical means.*

**Keywords:** spinel,  $MgAl_2O_4$ , solid-state synthesis, kinetics, mechanical activation, corundum, alumina, periclase, mechanical activation, infrared spectra, xerogel combustion product

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## INTRODUCTION

In recent decades, spinels have become widespread in various areas of material production, including ceramics, refractory products, catalysts, pigments, a variety of composites, and more.

The magnesia spinel (MS)  $MgAl_2O_4$  is the most well-known of this group, having the general formula  $M^II M^III_2O_4$ , where  $M^II$  and  $M^III$  are divalent and trivalent metal cations, respectively. MS has a unique set of physicochemical and operational characteristics: high melting point (~2135 °C), significant density (3.584 g/cm<sup>3</sup>) [1], high hardness (8 on the Mohs scale), mechanical strength [2, 3], which remains significant at high temperatures, low dielectric losses, good heat resistance, chemical and radiation resistance, low coefficient of thermal expansion, excellent resistance to acids and alkalis [4], etc. It should also be noted that, unlike many other refractory systems, the eutectics in the  $MgO-Al_2O_3$  system are also high-temperature (> 1900 °C).

MS is now widely used for the production of a variety of ceramic and refractory products [5-11].  $MgAl_2O_4$  spinel is incorporated in refractories to increase the high temperature durability in high demanding applications such as purging plugs or impact pads [12]. The MS use is also based on its optical properties [3, 13-16] and photoactivity [17, 18]. Luminescent materials [19], sensors [20], catalysts [21, 22] are made from MS. Spinel sol has been proposed as a binder in alumina refractories (castable) [23] and silicon carbide ceramics [24].

In this regard, it is understandable that interest in this compound has not decreased for more than half a century. The ever-increasing demand for this product stimulates the development of new and improvement of known methods for its production, such as ceramic synthesis [25], coprecipitation [26-29], sol-gel method [30-35], decomposition of double hydroxides [36], combustion [37-44], high-temperature self-propagating synthesis [45, 46], etc.

In industry, spinel is produced by solid-phase sintering or melting of mixtures of industrial powders [8,25, 47].

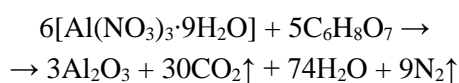
Corundum ( $\alpha$ - $\text{Al}_2\text{O}_3$ ) and periclase ( $\text{MgO}$ ) are known to be thermodynamically and kinetically stable, and therefore have low reactivity. In order to intensify the interaction of oxides, corundum and periclase are replaced by more active forms [48-52], and microwave [53-55] or mechanical [25, 55-59] treatment is used. Mechanical activation is a very common method of increasing the reactivity of various substances and materials, including for the spinel production. Most often, mechanochemical reactions are carried out in high-energy planetary impact-abrasion mills. The abrasion effect on solid-phase processes remains poorly understood [25, 60, 61].

The aim of this study was to compare the reactivity of alumina precursors by studying the kinetics of  $\text{MgAl}_2\text{O}_4$  solid-phase synthesis after treatment of initial oxides in activators. For the first time, alumina powder obtained by xerogel combustion was taken as one of the sources of  $\text{Al}_2\text{O}_3$ .

#### MATERIALS AND EXPERIMENTS

Industrial powders of sintered periclase with a 5-15  $\mu\text{m}$  fraction content of at least 70%, fused corundum (at least 80% of particles with a size of 20-25  $\mu\text{m}$ , specific surface area according to BET 0.45  $\text{m}^2/\text{g}$ ), metallurgical alumina grade G-00 (GOST 30558-98), non-metallurgical alumina grade GK (GOST 30559-98), reactive alumina (RG; TU 6-09-3916-75) with a specific surface area of 0.50; 0.71; 105  $\text{m}^2/\text{g}$ , respectively, were used.

To obtain alumina powder by combustion, aluminium nitrate crystal hydrate  $\text{Al}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$  was taken as an oxidizer and citric acid  $\text{C}_6\text{H}_8\text{O}_7$  as a fuel in a stoichiometric ratio according to the following reaction equation:



Concentrated solutions of nitrate and acid were mixed to form a transparent gel, from which a xerogel was obtained by drying at 70-90  $^\circ\text{C}$ . The latter was ignited by heating on an electric stove at a temperature of 200-300  $^\circ\text{C}$ . The combustion product was pounded in a mortar; part of it was annealed at 500  $^\circ\text{C}$ .

Mechanical processing was carried out in two types of laboratory mills: in a ball-ring mill (BRM), operating on the principle of abrasion; in the AGO-2 planetary mill (PM), the action of which includes an impact component.

IR spectra of samples were obtained on the device "Avatar 360-FT-IR" (company "Nicolet") in the region of 500-4000  $\text{cm}^{-1}$ .

The specific surface area of powders was determined on the Sorby M device using nitrogen adsorption-desorption data and the BET method.

Scanning electron microscopy (Vega 3 SBH) was used to describe the morphology of the crystals.

Charges were prepared by simple mixing the components; mixing oxides, one of which has been pre-treated in a mill; co-processing of components in a mill. The initial substances were taken in quantities corresponding to their stoichiometric ratio in spinel. Tablets were pressed (200 MPa) from the reaction mixtures using a 5% solution of polyacrylic acid. The samples were then fired at 1300  $^\circ\text{C}$ .

The yield of  $\text{MgAl}_2\text{O}_4$  was determined by quantitative X-ray phase analysis ( $d = 0.143$  and  $0.105$  nm) using the DRON-6 diffractometer [62]. For control, thermogravimetric analysis was performed (by weighing the insoluble residue in concentrated hydrochloric acid).

#### RESULTS

It is known that industrial oxide objects often contain some adsorbed water, so at high temperatures a partial change in the periclase component ( $\text{MgO}$ ) due to hydration is possible. The presence of hydrate water,  $\text{OH}^-$  and  $\text{CO}_3^{2-}$  groups can be conveniently monitored using IR spectroscopy. The IR spectra of spinel precursors, namely periclase and corundum after mechanical treatment, were analyzed (Fig. 1).

The initial fused corundum powder KP (curves 1 in Fig. 1, a, b) contains  $\text{OH}$  groups (valence oscillations of isolated  $\text{OH}^-$  groups at  $\sim 3700$   $\text{cm}^{-1}$ ) and  $\text{CO}_3^{2-}$  groups (valence group oscillations in the region of  $\sim 1630$  and  $1420$ - $1480$   $\text{cm}^{-1}$ ), and the carbonate groups are chemisorbed in positions that are energetically unequivalent, which is confirmed by splitting a wide single band into a clearly distinguishable doublet of sharp peaks of  $1420$  and  $1480$   $\text{cm}^{-1}$ . There is a small amount of adsorbed water (a wide band of  $3300$ - $3600$   $\text{cm}^{-1}$  – valence fluctuations of the  $\text{OH}$ -groups of molecular water). In the short-wave region, there is a strong absorption in the region of  $760$ ,  $750$ ,  $670$ ,  $680$ - $500$ ,  $530$ ,  $400$   $\text{cm}^{-1}$ , characteristic of  $\text{AlO}_6$  octahedra.

Periclase (curves 2 in Fig. 1 a, b) is characterized by the absence of peaks for  $\text{OH}$ -groups, there are very faint bands of  $\text{CO}_3^{2-}$  groups; there is adsorbed water. In the short-wavelength region, moderate absorption is in the region of  $450$ - $750$   $\text{cm}^{-1}$ , corresponding to a highly extended band with a maximum absorption of  $\sim 580$ - $590$   $\text{cm}^{-1}$  (valence oscillations of  $\text{MgO}_6$  groups).

After treatment in the planetary mill, changes in the corundum spectrum (Fig. 1 a, curve 3) indicate the destruction of all compounds formed as a result of

chemisorption. A strong amorphization becomes apparent, confirmed by an absorption decrease in the entire range of wave numbers and by the absence of pronounced absorption bands. Under the same conditions, initially inactive periclase is activated, which is accompanied by the addition of  $\text{OH}^-$  and  $\text{CO}_3^{2-}$  due to chemisorption (Fig. 1 a, curve 4).

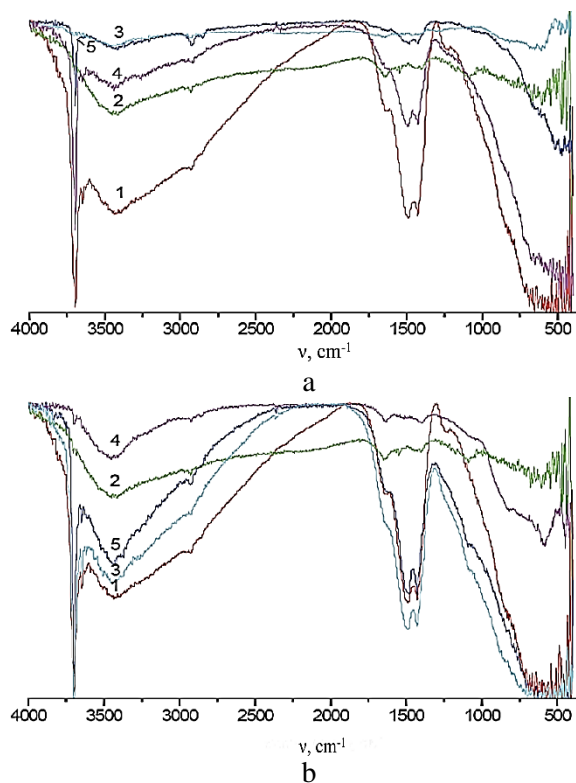


Fig. 1. IR spectra of initial (1,2) and processed (3,4,5) oxides in planetary (a) and ball-ring mill (b): 1,3 – corundum; 2,4 – periclase; 5 – corundum + periclase. Machining time, min: a – 20; b – 5  
 Рис. 1. ИК спектры исходных (1,2) и обработанных (3,4,5) в планетарной (а) и шаро-кольцевой мельнице (б) оксидов: 1,3 – корунд; 2,4 – периклаз; 5 – корунд + периклаз. Время механической обработки, мин: а – 20; б – 5

Abrasive treatment of corundum in air leads to a slight increase in the content of  $\text{OH}^-$  and  $\text{CO}_3^{2-}$  (Fig. 1 b, curve 3). An increase in absorption in the region of  $750\text{--}900\text{ cm}^{-1}$  may indicate the appearance of some aluminum in the tetrahedral coordination ( $\text{AlO}_4$ ). As a result of similar treatment of periclase, a noticeable change in the nature of absorption in the short-wavelength region is observed (Fig. 1 b, curve 4): the diffuse band corresponding to the valence fluctuations of the  $\text{Mg-O}$  bonds turns into a clear line with a maximum of  $\sim 600\text{ cm}^{-1}$ , which may apparently indicate some ordering of the structure.

The IR spectra of oxide mixtures after treatment are intermediate (Fig. 1 a, b, curves 5). No new absorption bands corresponding to the appearance of

the spinel phase were detected. In this way, the short pre-machining of the initial components not only activates them, but also contributes to the destruction of the adsorption compounds already in the preparation phase.

X-ray phase analysis of the xerogel combustion product (Fig. 2) indicates its high X-ray amorphousness, and therefore high chemical activity.

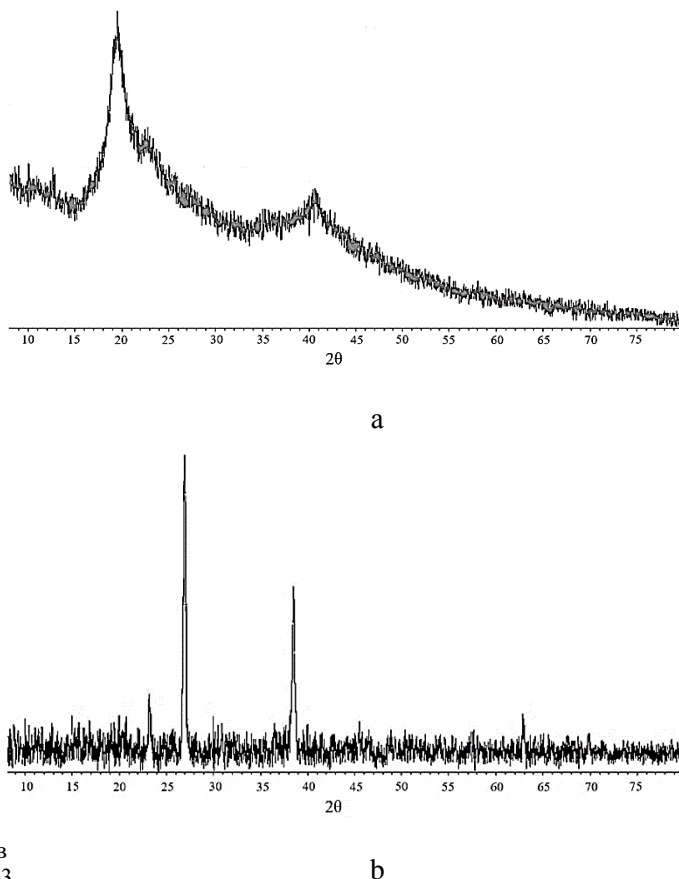


Fig. 2. XRD spectra of the xerogel combustion product from aluminum nitrate and citric acid at the ratio of the initial components 1:1 (a) and 1:1.25 (b) without annealing

Рис. 2. Дифрактограммы продукта горения ксерогеля из нитрата алюминия и лимонной кислоты при соотношении исходных компонентов 1:1 (а) и 1:1,25 (б) без отжига

When corundum was replaced by the xerogel combustion product for the spinel formation reaction, the relative content of the initial oxides, especially  $\text{Al}_2\text{O}_3$ , decreased (Fig. 3), indicating a more complete binding of the reactants.

To determine the rate constants of the spinel formation reaction, the yield of  $\text{MgAl}_2\text{O}_4$  during the roasting process at  $1300\text{ }^\circ\text{C}$  was determined (Fig. 4). The designations of the series of experiments are given in Table 1.

Simple mixing of the components resulted in a relatively small amount of product (up to  $\sim 40\%$ ,

Fig. 4, curve 1). If one of the reagents is pre-treated mechanically (Fig. 4, curves 2-3), the product yield is significantly increased (up to 60-80%) due to the solid phase activation by increasing the structure defectiveness. The greater influence of periclase processing compared to corundum was explained earlier [25] by the peculiarities of the periclase crystal lattice of the halite type, which has many sliding planes, in which dislocations are more easily formed. Co-activation (Fig. 4, curve 4) leads to the maximum result, since it contributes to a significant increase in the reactants contact surface. The efficiency of the ball-ring mill is

noticeably lower (Fig. 4, curve 5) due to its lower energy intensity.

If corundum powder KP is replaced by another industrial source of alumina (GK, G-00), the spinel yield remains almost the same (Fig. 4, curves 6, 7); only for reactive alumina the synthesis productivity increase slightly (Fig. 4, curve 8). This can be attributed to the fact that the specific surface area of the initial powders KP, GC and G-00 differs slightly: 0.45; 0.50; 0.71 m<sup>2</sup>/g, respectively. Their phase composition is represented by high-temperature corundum  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> (Fig. 5 a-c).

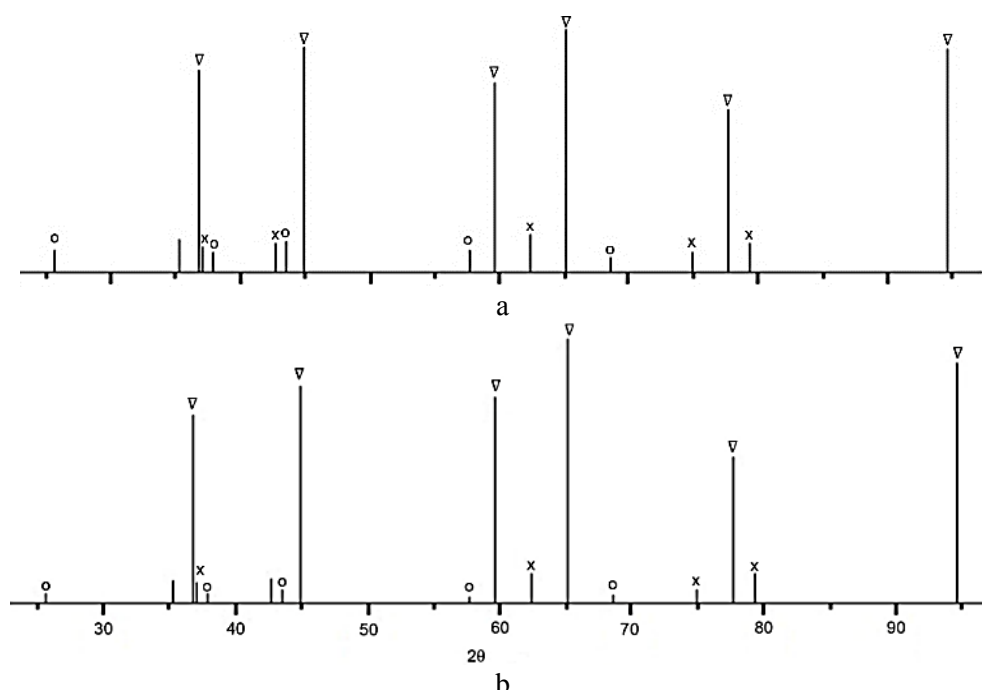


Fig. 3. Bar diagrams for machined and fired (1300 °C) mixture of periclase and corundum (a); for the fired mixture of periclase and the combustion product of aluminium nitrate with citric acid (1:1) without annealing (b). Designations: o –  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>; x – MgO;

∇ – MgAl<sub>2</sub>O<sub>4</sub> Обозначения: o –  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>; x – MgO; ∇ – MgAl<sub>2</sub>O<sub>4</sub>

Рис. 3. Штрих-диаграммы для механически обработанной и обожженной (1300 °C) смеси периклаза и корунда (а); для обожженной смеси периклаза и продукта горения нитрата алюминия с лимонной кислотой (1:1) без отжига (б). Обозначения: o –  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>; x – MgO; ∇ – MgAl<sub>2</sub>O<sub>4</sub>

Table 1

**Designations of a series of experiments to study the kinetics of spinel formation**

**Таблица 1. Обозначения серии экспериментов по изучению кинетики образования шпинели**

№	Method for preparing reaction mixture of initial components	Обозначение
1	A mixture made by a simple mixture of periclase and corundum	P+C
2	Mixture of periclase and corundum treated in PM	P+C(PM)
3	Mixture of PM-treated periclase and corundum	P(PM)+C
4	Mixture of periclase and corundum treated in PM	(P+C)(PM)
5	Mixture of periclase and corundum processed in BRM	(P+C)(BRM)
6	Mixture of periclase and alumina HA processed in PM	(P+GK)(PM)
7	Mixture of periclase and alumina G-00 treated in PM	(P+G-00)(PM)
8	Mixture of periclase and reactive alumina treated in PM	(P+RG)(PM)
9	Mixture of xerogel combustion product and periclase	XCP+P

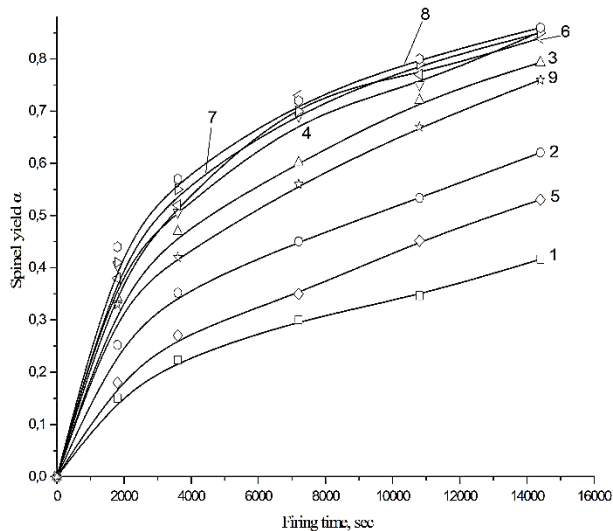


Fig. 4. Spinel yield from mixtures: P+C (1); P+C(PM) (2); P(PM)+C (3); (P+C)(PM) (4); (P+C)(BRM) (5); (P+GK)(PM) (6); (P+G-00)(PM) (7); (P+RG)(PM) (8); XCP+P (9). Firing temperature is 1300 °C

Рис. 4. Выход шпинели из смесей: П+К (1); П+К(ПМ) (2); П(ПМ)+К (3); (П+К)(ПМ) (4); (П+К)(ШКМ) (5); (П+ГК)(ПМ) (6); (П+Г00)(ПМ) (7); (П+РГ)(ПМ) (8); ПГК+П (9). Температура обжига 1300 °C

Reactive alumina is represented by a mixture of transition forms of alumina partially superimposed on each other (Fig. 5 d). They are more reactive, in addition, its specific surface area is significantly higher, which is also confirmed by microphotographs (Fig. 6

d-e). All this leads to a certain increase in the  $MgAl_2O_4$  yield (Fig. 4, curve 8).

Of interest is curve 9 in Fig. 4, relating to the use of the xerogel combustion product from aluminium nitrate and citric acid as a source of  $Al_2O_3$ . Even in the absence of mechanical activation of the mixture components, it is possible to achieve a high spinel yield (up to 80%). Apparently, the formation of alumina under the conditions of the release of a huge volume of gaseous products (10 moles of  $CO_2$ , 24.6 moles of  $H_2O$ , 3 moles of  $N_2$  per 1 mol of  $Al_2O_3$ ) contributes to the breaking of chemical bonds and, as a result, leads to the absence of a well-formed crystal lattice of alumina, which increases the reactivity of this component.

To calculate the effective velocity constants, the Ginstling-Brownstein equation was used, which had previously [25] shown to be suitable for processing similar data:

$$1 - \frac{2}{3}\alpha - (1 - \alpha)^{\frac{2}{3}} = K\tau, \quad (1)$$

where  $\alpha$  is the spinel yield (degree of transformation),  $\leq 1$ ;  $K$  is the effective reaction rate constant,  $sec^{-1}$ ;  $\tau$  – time, sec.

In the coordinates of this equation, dependencies with a confidence coefficient of the linear approximation  $R^2$  of at least 0.95 were obtained (Fig. 7). Their inclination tangent yielded effective reaction rate constants (Table 2).

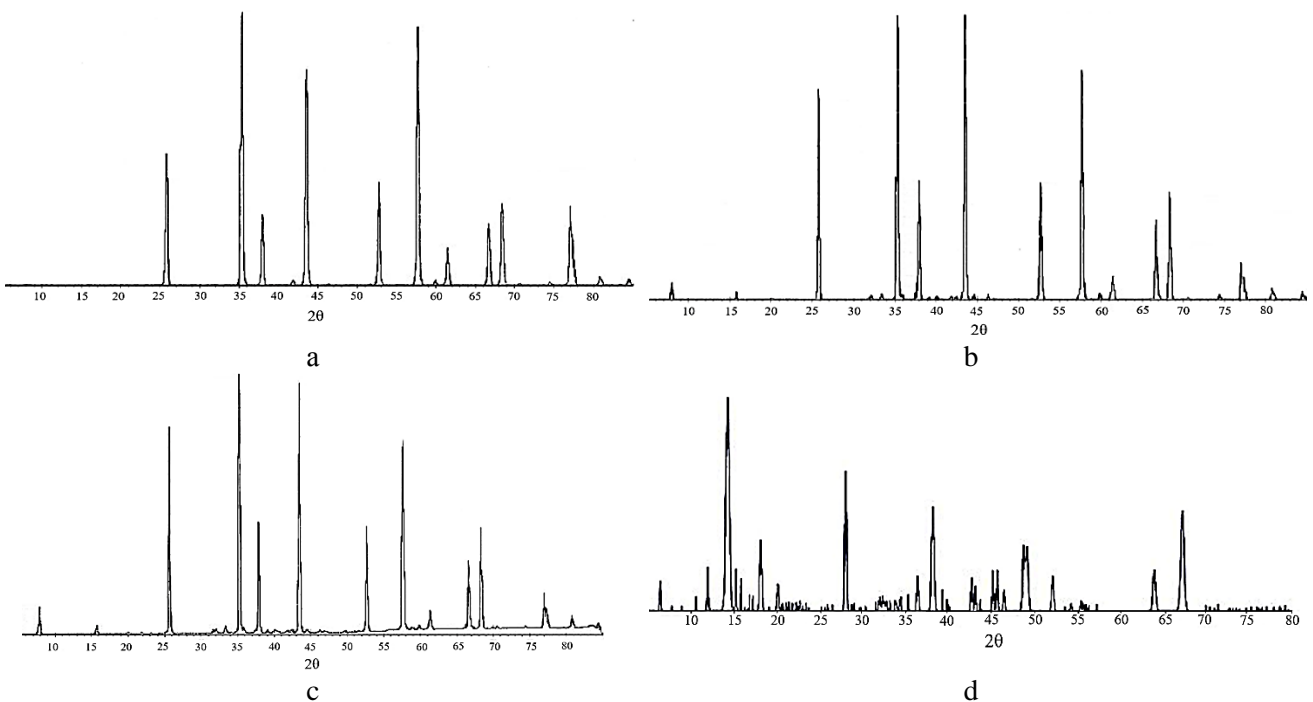


Fig. 5. XRD spectra of KP (a), GC (b), G-00 (c), RG (d)  
Рис. 5. Дифрактограммы КП (a), ГК (b), Г-00 (c), РГ (d)



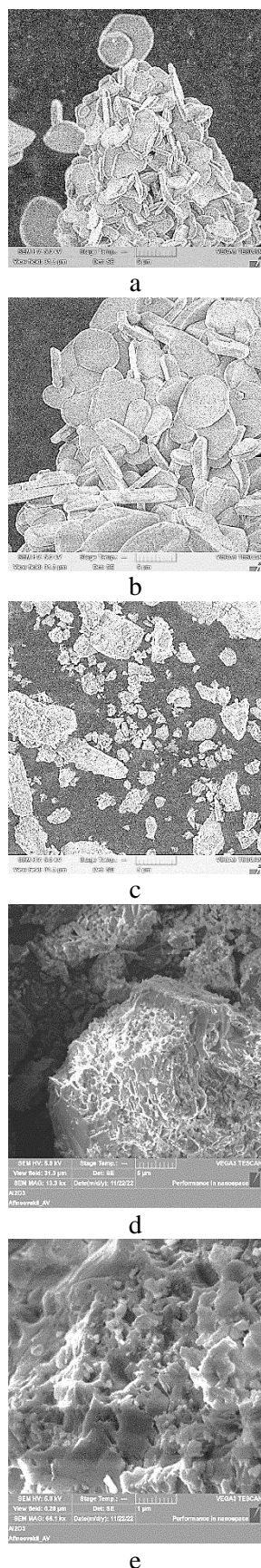


Fig. 6. SEM photos of KP (a), GC (b), G-00 (c), RG (d, e)  
Рис. 6. СЭМ-фотографии КП (а), ГК (b), Г-00 (c), РГ (d, e)

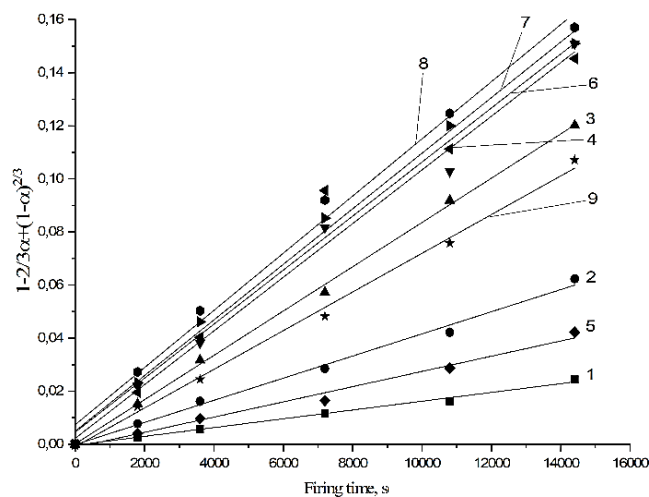


Fig. 7. Dependencies in coordinates of the Ginstling-Brownstein equation for mixtures: P+C (1); P+C(PM) (2); P(PM)+C (3); (P+C)(PM) (4); (P+C)(BRM) (5); (P+GK)(PM) (6); (P+G-00)(PM) (7); (P+RG)(PM) (8); XCP+P (9). Firing temperature is 1300 °C

Рис. 7. Зависимости в координатах уравнения Гинстлинга-Броунштейна для смесей: П+К (1); П+К(ПМ) (2); П(ПМ)+К (3); (П+К)(ПМ) (4); (П+К)(ШКМ) (5); (П+ГК)(ПМ) (6); (П+Г00)(ПМ) (7); (П+РГ)(ПМ) (8); ПГК+П (9). Температура обжига 1300 °C

Table 2

Effective rate constants of  $MgAl_2O_4$  formation (1300 °C)

Таблица 2. Эффективные константы скорости реакции образования  $MgAl_2O_4$  (1300 °C)

Reaction mixture designation	Rate constants $K \cdot 10^6, s^{-1}$
P+C	$1.7 \pm 0.1$
P+C(PM)	$4.1 \pm 0.2$
P(PM)+C	$8.3 \pm 0.2$
(P+C)(PM)	$10.1 \pm 0.5$
(P+C)(BRM)	$2.9 \pm 0.2$
(P+GC)(PM)	$10.2 \pm 0.8$
(P+G-00)(PM)	$10.5 \pm 0.4$
(P+RG)(PM)	$10.8 \pm 0.5$
XCP+P	$7.3 \pm 0.2$

The obtained constant values indicate a greater efficiency of impact processing. Co-mechanical activation of reagents in a planetary mill can increase the reaction rate by ~ 5 times, while abrasion in a ball-ring mill is only 1.7 times. Pretreatment of one component of the charge in PM is the most expedient for periclase, since it allows accelerating the reaction by ~ 5 times, and is also energetically and technologically the most rational due to the treatment of only one component of reduced hardness [25]. The use of xerogel combustion product in the synthesis of spinel is very effective, because it accelerates the process by ~4 times even without preliminary activation by mechanical means.

## CONCLUSION

A comparison of the efficiency of solid-phase synthesis of magnesia spinel  $MgAl_2O_4$  from magnesia and alumina using a mechanical activation of initial reagents and industrial powders (corundum powder KP, alumina: metallurgical GK, non-metallurgical G-00, reactive RG) and xerogel combustion product from aluminum nitrate and citric acid as a source of  $Al_2O_3$  was carried out. It is shown that in the presence of the combustion product, the spinel yield increases significantly, approaching the synthesis productivity after preliminary mechanical treatment of the reaction mixture containing one of the industrial alumina powders.

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The authors declare the absence a conflict of interest warranting disclosure in this article.

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